

Phase Relationship in System $3\text{CaO} \cdot \text{P}_2\text{O}_5$ - MgO - SiO_2

By Toshiyuki SATA

(Received July 6, 1960)

The author already presented in 1958 the result of a study on system $3\text{CaO} \cdot \text{P}_2\text{O}_5$ - MgO - SiO_2 - SiO_2 in this Bulletin¹⁾. This is a partial system at high silica portion in the three-component system $3\text{CaO} \cdot \text{P}_2\text{O}_5$ - MgO - SiO_2 . The present report concerns a basic portion in the system $3\text{CaO} \cdot \text{P}_2\text{O}_5$ - MgO - SiO_2 , and its phase relationships are important for the study of a composition of fused phosphate fertilizer²⁾ and a corrosion problem of magnesia-containing basic refractories against the molten fertilizer. This report consists of four parts; system $3\text{CaO} \cdot \text{P}_2\text{O}_5$ - MgO , system $3\text{CaO} \cdot \text{P}_2\text{O}_5$ - 2MgO - SiO_2 , system $3\text{CaO} \cdot \text{P}_2\text{O}_5$ - MgO - 2MgO - SiO_2 , and the synthesis of these to system $3\text{CaO} \cdot \text{P}_2\text{O}_5$ - MgO - SiO_2 . Phase equilibria studies in these systems have never been published in any paper.

Experimental

Tricalcium phosphate ($3\text{CaO} \cdot \text{P}_2\text{O}_5$), forsterite ($2\text{MgO} \cdot \text{SiO}_2$), and magnesia (MgO) were used as raw materials. The first was prepared by the reaction of calcium carbonate with phosphoric acid at 1350°C as mentioned in the previous paper¹⁾. The second was prepared by calcining the stoichiometric mixture of magnesia and silica constituents at 1300°C . The magnesia and silica fired from 1200 to 1300°C were used as raw materials respectively. After weighing, the batches were thoroughly mixed in an agate mortar with some methyl alcohol until they became dry. They were heated in a platinum crucible or in a platinum envelope above their melting points. Then they were rapidly cooled in air or water. For batches having the melting point above 1600°C , perfect

classification was impossible. These mixing, heating, and quenching processes were repeated once more.

The silicon carbide resistor furnace with an inner diameter of 25 mm. was used for this study. The sample in a platinum envelope (8×8 mm.) was suspended by a platinum wire (0.5 mm. in diameter, 200 mm. in length) at the center of the furnace tube, and quenched together with the wire in water after holding for some minutes at constant temperature ($\pm 1^\circ\text{C}$). For measuring the temperature, two kinds of thermocouples (Pt-PtRh 10% and PtRh 5%-PtRh 20%) were used. These were calibrated with the melting point of silver, copper, diopside, and palladium.

The quenched samples were powdered and examined by a polarizing microscope and by an X-ray diffractometer. Pure silicon metal was used as a standard material in X-ray examination, its accuracy being $\pm 0.003 \text{ \AA}$.

Results and Discussion

System $3\text{CaO} \cdot \text{P}_2\text{O}_5$ - MgO .—In the first place, specimens slowly cooled from 1650°C were examined in their microstructure. The specimen containing 3.20% of MgO consisted of a primary crystal part of β -tricalcium phosphate and an eutectic part which is a turbid phase containing periclase spots surrounded by tricalcium phosphate matrix. The specimen containing 5.28% of MgO revealed the eutectic structure only. The specimen containing 7.98% of MgO consisted of the eutectic part and large periclase grains. The softening temperatures of small cones of these three compositions were measured on platinum plates in the usual way. It was found that their temperatures decreased in order of contents of MgO ranging from 1620 to 1530°C .

1) T. Sata, This Bulletin, 31, 408 (1958).

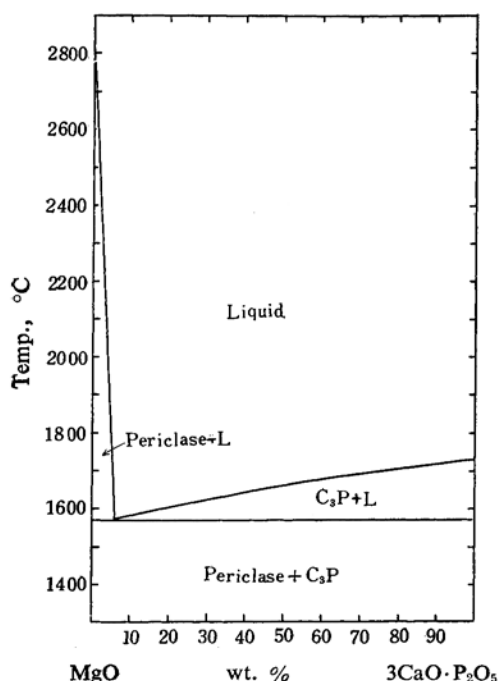
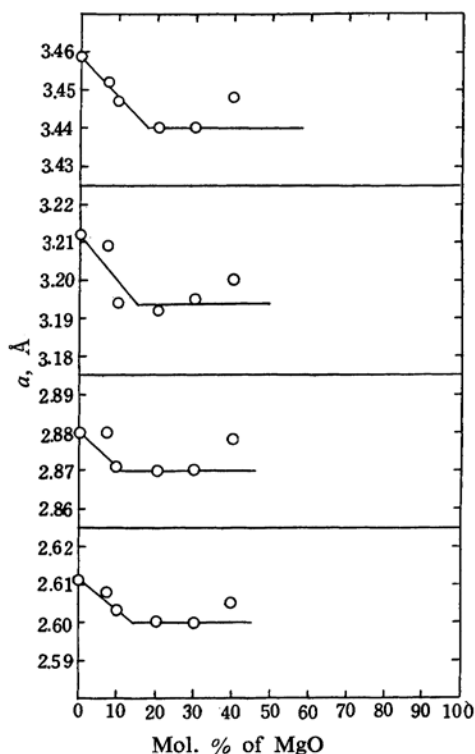
2) T. Sata and R. Kiyoura, J. Chem. Soc. Japan, Ind. Chem. Sec. (Kogyo Kagaku Zasshi), 62, 63 (1959).

TABLE I. QUENCHING DATA IN THE SYSTEM $3\text{CaO} \cdot \text{P}_2\text{O}_5$ -MgO

| No. | Composition | | | Holding | | Phases present |
|-----|-------------------------------|------|---------------|-------------|--------------|---|
| | C_3P wt. % | MgO | MgO mol. % | temp. °C | time min. | |
| 193 | 99 | 1.0 | 7.2 | 1470 | 60 | β - C_3P , α - C_3P |
| 19 | 98.58 | 1.42 | 10.0 | 1450 | 60 | β - C_3P |
| 20 | 96.8 | 3.2 | 20.0 | 1568 | 15 | β - C_3P , small α - C_3P , periclase, glass |
| | | | | 1450 | 60 | β - C_3P , periclase |
| 21 | 94.7 | 5.3 | 30.0 | 1568 | 15 | β - C_3P , periclase |
| 22 | 92.1 | 7.9 | 40.0 | 1568 | 15 | β - C_3P , periclase |
| | | | | 1555 | 10 | " |
| | | | | 1524 | 10 | " |

The subsequent quenching runs, as listed in Table I, indicated that the specimen containing 3.20% of MgO, quenched from 1568°C, consisted of tricalcium phosphate (β and small amount of α), periclase, and glass. It is estimated from these results that this $3\text{CaO} \cdot \text{P}_2\text{O}_5$ -MgO system is a binary one and its eutectic occurred at a composition of 4~7% of MgO and at a temperature near $1570 \pm 20^\circ\text{C}$, as shown in Fig. 1.

It is noted from Fig. 2, in which some X-ray diffraction lines of β -tricalcium phosphate quenched from 1450°C are plotted, that 15 ± 3 mol. % of MgO is apparently dissolved into β -tricalcium phosphate as a solid solution. Another fact is that the transition temperature of β - α modification is considerably elevated nearly to 1550°C with increasing magnesia contents. These facts concerning tricalcium

Fig. 1. The system $3\text{CaO} \cdot \text{P}_2\text{O}_5$ -MgO.Fig. 2. The change in spacings of β -tricalcium phosphate at 1450°C in the system $3\text{CaO} \cdot \text{P}_2\text{O}_5$ -MgO.

phosphate are difficult to discuss further without direct examination by a high-temperature X-ray technique. There is a new presentation of super α - $3\text{CaO} \cdot \text{P}_2\text{O}_5$ phase by Nurse et al.³⁾

System $3\text{CaO} \cdot \text{P}_2\text{O}_5$ - $2\text{MgO} \cdot \text{SiO}_2$.—Fasting and Haroldsen⁴⁾ have made X-ray tests of specimens in this system fired at 1000 ~ 1500°C , and Bobrownicki and Swiecki⁵⁾ have measured

3) R. W. Nurse, J. H. Welch and W. Gutt, *J. Chem. Soc.*, 1959, 1077.

4) J. P. Fasting and H. Haroldsen, *Tid. Kjem., Bergvesen Met.*, 3, 48 (1943); *Chem. Abstr.*, 39, 13419 (1945).

5) W. Bobrownicki and Z. Swiecki, *Bull. Acad. Polon. Sci., Classe III*, 3, 232 (1955); *Chem. Abstr.*, 49, 15419 (1955).

TABLE II. QUENCHING DATA IN THE SYSTEM $3\text{CaO} \cdot \text{P}_2\text{O}_5$ - $2\text{MgO} \cdot \text{SiO}_2$

| No. | Composition | | Holding | | Phases present |
|-----|-------------------------------|----------------------|-------------|--------------|--|
| | C_3P wt. % | M_2S | temp. °C | time min. | |
| 191 | 98 | 2 | 1342 | 60 | β - C_3P only |
| 75 | 95 | 5 | 1370 | 60 | β - C_3P , gl. |
| | | | 1340 | 40 | β - C_3P |
| | | | 1290 | 120 | β - C_3P |
| | | | 1242 | 180 | β - C_3P , faint forst. |
| | | | 1342 | 60 | β - C_3P only |
| 192 | 93 | 7 | 1517 | 30 | α - C_3P , β - C_3P , gl. |
| 71 | 90 | 10 | 1467 | 60 | α - C_3P , β - C_3P , gl. |
| | | | 1442 | 50 | β - C_3P , α - C_3P , gl. |
| | | | 1392 | 60 | β - C_3P , gl. |
| | | | 1372 | 540 | β - C_3P , gl. |
| | | | 1340 | 40 | β - C_3P |
| | | | 1312 | 40 | β - C_3P |
| | | | 1290 | 40 | β - C_3P , forst. |
| | | | 1545 | 20 | Ng., gl. |
| | | | 1442 | 20 | Ng., gl. |
| | | | 1392 | 60 | Ng., β - C_3P , gl. |
| 70 | 80 | 20 | 1372 | 60 | β - C_3P , Ng., gl. |
| | | | 1357 | 40 | β - C_3P , gl. |
| | | | 1352 | 300 | β - C_3P , forst., small gl. |
| | | | 1342 | 60 | β - C_3P , forst. |
| | | | 1575 | 15 | gl. |
| | | | 1566 | 15 | Ng., gl. |
| | | | 1515 | 15 | Ng., gl. |
| | | | 1374 | 40 | β - C_3P , Ng., gl. |
| | | | 1352 | 40 | β - C_3P , forst. |
| | | | 1548 | 15 | Ng., gl. |
| 73 | 72.5 | 27.5 | 1528 | 15 | Ng., gl. |
| | | | 1442 | 60 | Ng., gl. |
| | | | 1382 | 40 | β - C_3P , Ng., gl. |
| | | | 1372 | 40 | β - C_3P , Ng., forst., gl. |
| | | | 1365 | 50 | β - C_3P , forst., small gl. |
| | | | 1352 | 40 | β - C_3P , forst. |
| | | | 1398 | 40 | gl. |
| | | | 1392 | 40 | β - C_3P , Ng., faint forst., gl. |
| | | | 1372 | 40 | β - C_3P , Ng., forst., gl. |
| | | | 1347 | 40 | β - C_3P , forst. |
| 74 | 67.5 | 32.5 | 1432 | 60 | gl. |
| | | | 1407 | 50 | β - C_3P , small forst., gl. |
| | | | 1374 | 40 | β - C_3P , forst. |
| 67 | 60 | 40 | 1465 | 40 | gl. |
| | | | 1458 | 20 | forst. gl. |
| | | | 1452 | 40 | Ng., forst., gl. |
| | | | 1400 | 60 | Ng., small forst. |
| | | | 1382 | 50 | Ng., forst. |
| | | | 1342 | 60 | β - C_3P , forst. |
| | | | 1300 | 60 | β - C_3P , forst. |
| | | | 1537 | 20 | gl. |
| 68 | 50 | 50 | 1522 | 20 | forst., gl. |
| | | | 1507 | 20 | forst., gl. |
| | | | 1442 | 60 | Ng., forst., gl. |
| | | | 1442 | 60 | Ng., forst., gl. |

Abbreviations: $\text{C}_3\text{P} = 3\text{CaO} \cdot \text{P}_2\text{O}_5$, $\text{M}_2\text{S} = 2\text{MgO} \cdot \text{SiO}_2$, Ng. = nagelschmidite
forst. = forsterite, gl. = glass.

the softening points of this system. But the phase relationships of this system have not been presented.

Results of the quenching runs of this two-component system are summarized in Table II, from which a preliminary phase diagram of this system was obtained as shown in Fig. 3. Nagelschmidtite⁶⁾ (which was identified by a hexagonal X-ray pattern similar to that in $3\text{CaO}\cdot\text{P}_2\text{O}_5$ - $2\text{CaO}\cdot\text{SiO}_2$ system) appeared over a wide range of this system above a temperature of 1350°C , and was crystallized primarily in the composition range of $3\text{CaO}\cdot\text{P}_2\text{O}_5$ side.

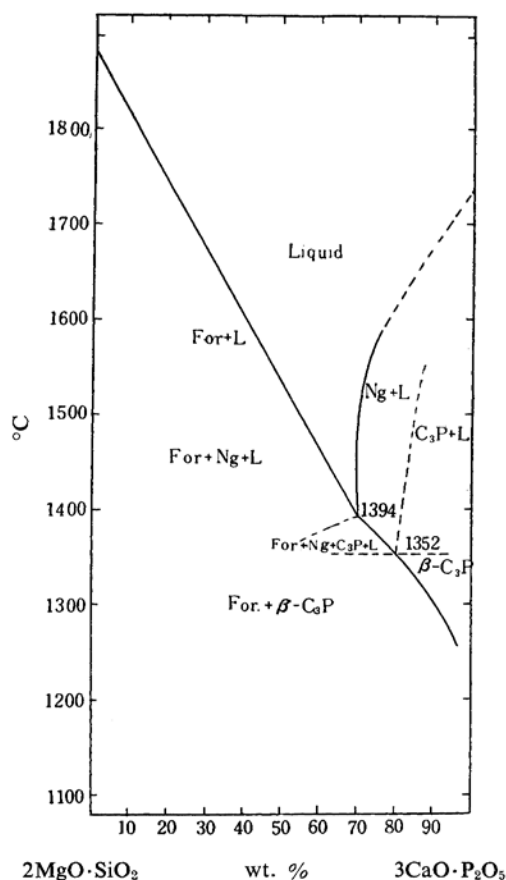


Fig. 3. The system $3\text{CaO}\cdot\text{P}_2\text{O}_5$ - $2\text{MgO}\cdot\text{SiO}_2$.

The liquidus line falls very steeply to a temperature minimum at 1394°C and a composition of 70% $3\text{CaO}\cdot\text{P}_2\text{O}_5$ and 30% $2\text{MgO}\cdot\text{SiO}_2$. Below this temperature 1394°C , the secondary crystallization temperature of forsterite from nagelschmidtite and liquid decreases. Below 1352°C , nagelschmidtite and liquid disappeared, and forsterite and β -tricalcium phosphate phase only appeared. The meaning of this disappearance temperature 1352°C is

not clear now. β -Tricalcium phosphate seems to contain magnesia component, judging from the fact that the crystallization temperature of forsterite from β -tricalcium phosphate phase descends with increasing contents of $3\text{CaO}\cdot\text{P}_2\text{O}_5$, and d -spacings of β -tricalcium phosphates in their X-ray patterns decrease till the content of 15 mol. % (7.4 weight %) of $2\text{MgO}\cdot\text{SiO}_2$ is attained, as shown in Fig. 4. Below about 1200°C , this two-component system is a binary one consisting of forsterite and β -tricalcium phosphate.

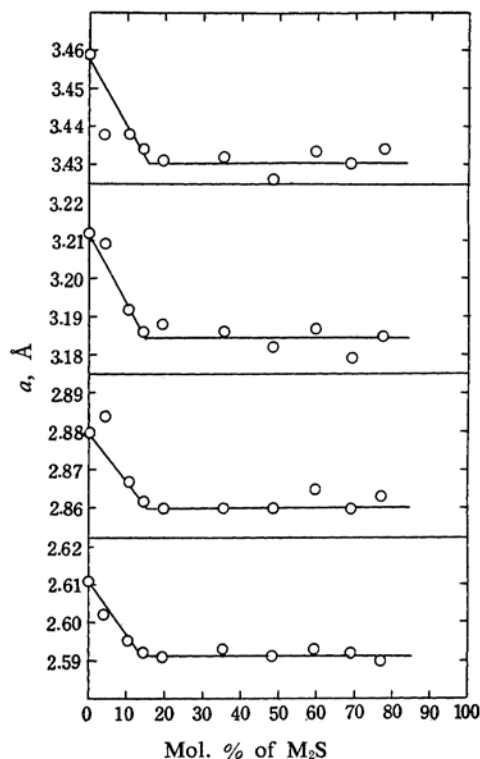


Fig. 4. The change in spacings of β -tricalcium phosphate in the system $3\text{CaO}\cdot\text{P}_2\text{O}_5$ - $2\text{MgO}\cdot\text{SiO}_2$.

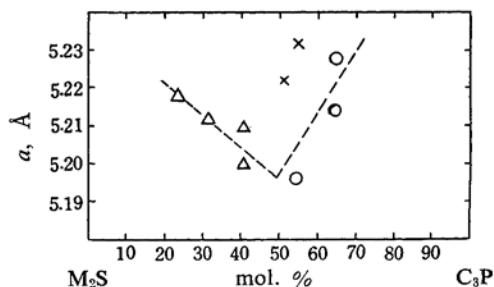


Fig. 5. The change in lattice constants of nagelschmidtite in the system $3\text{CaO}\cdot\text{P}_2\text{O}_5$ - $2\text{MgO}\cdot\text{SiO}_2$ at $1370\sim1490^\circ\text{C}$.

○ Ng+L, △ Ng+forst.+L,
× Ng+ β - C_3P +L

6) G. Nagelschmidt, *J. Chem. Soc.*, 1937, 865.

When the lattice constants a of hexagonal nagelschmidite are calculated from their (110) peak and plotted against molar compositions, a valley-type change are obtained, whose bottom may occur at the composition of temperature minimum of liquidus as shown in Fig. 5.

From the results mentioned above, this system was found to be complex, owing to the formation of nagelschmidite above 1350°C , below which it is considered to be binary.

System $3\text{CaO} \cdot \text{P}_2\text{O}_5$ - MgO - $2\text{MgO} \cdot \text{SiO}_2$. — Although all of the specimens of this system slow-cooled to room temperature consisted of β -tricalcium phosphate, periclase, and forsterite, the primary crystal in the area of high content of $3\text{CaO} \cdot \text{P}_2\text{O}_5$ was nagelschmidite, being the same as that mentioned in the system $3\text{CaO} \cdot \text{P}_2\text{O}_5$ - $2\text{MgO} \cdot \text{SiO}_2$. But it was found by the preliminary quenching test that a ternary eutectic among their three components seems to be present. This paragraph described results of quenching experiments made to decide the position of a three-component eutectic.

The twelve compositions, which are expected to locate near the eutectic, were taken on the two tie lines connecting No. 24 ($60\text{C}_3\text{P}$ - 40M) with No. 72 ($75\text{C}_3\text{P}$ - $25\text{M}_2\text{S}$) and No. 74 ($67.5\text{C}_3\text{P}$ - $32.5\text{M}_2\text{S}$) along the C_3P - M_2S join, as represented in Fig. 6. Results obtained are listed in Table III, from which a eutectic position was graphically decided to locate at the composition of 70% $3\text{CaO} \cdot \text{P}_2\text{O}_5$, 21% MgO and 9% SiO_2 , and at $1362 \pm 5^\circ\text{C}$.

This eutectic temperature is close to the transition point between nagelschmidite and β -tricalcium phosphate. It was, however, found that β -tricalcium phosphate was present at this eutectic temperature. The primary field of nagelschmidite was approximately estimated as a dotted boundary from the previous results of the investigation in the systems C_3P - MS and C_3P - M_2S . The facts that a small amount of periclase was present even in the specimens quenched above liquidus temperatures of all compositions, and that nagelschmidite was present even in the primary field of periclase or forsterite above eutectic

TABLE III. QUENCHING DATA IN THE SYSTEM $3\text{CaO} \cdot \text{P}_2\text{O}_5$ - MgO - $2\text{MgO} \cdot \text{SiO}_2$

| No. | Composition | | | | | Holding | | Phases present (peri.=periclase) |
|-----|-------------|----------------------|------|------|------|---------------------------|--------------|---|
| | wt. % | C_3P | M | S | | temp. $^\circ\text{C}$ | time min. | |
| | No. 74 | No. 24 | | | | | | |
| 78 | 95 | 5 | 67.1 | 19.7 | 13.2 | 1442 | 60 | gl. (faint peri.) |
| | | | | | | 1407 | 40 | small forst., peri., gl. |
| | | | | | | 1364 | 60 | β - C_3P , forst., peri. |
| 80 | 90 | 10 | 66.8 | 20.7 | 12.5 | 1442 | 60 | small peri., gl. |
| | | | | | | 1412 | 50 | small forst., gl. |
| | | | | | | 1392 | 50 | β - C_3P , forst., gl. |
| 81 | 85 | 15 | 66.4 | 21.8 | 11.8 | 1407 | 50 | small peri., gl. |
| | | | | | | 1382 | 50 | β - C_3P , small forst., gl. |
| 82 | 80 | 20 | 66.0 | 22.9 | 11.1 | 1422 | 40 | gl. |
| | | | | | | 1392 | 60 | peri., gl. |
| | | | | | | 1382 | 50 | β - C_3P , forst., gl. |
| 83 | 75 | 25 | 65.6 | 24.0 | 10.4 | 1390 | 50 | peri., gl. |
| | | | | | | 1382 | 50 | β - C_3P , forst., peri., gl. |
| 101 | 65 | 35 | 64.9 | 26.1 | 9.0 | 1382 | 40 | peri., gl. |
| | | | | | | 1372 | 40 | peri., β - C_3P , forst. |
| 102 | 55 | 45 | 64.2 | 28.2 | 7.6 | 1365 | 20 | peri., gl. |
| | | | | | | 1356 | 60 | peri., β - C_3P , forst. |
| 103 | 45 | 55 | 63.3 | 30.4 | 6.3 | 1375 | 40 | peri., gl. |
| | | | | | | 1356 | 40 | β - C_3P , forst., peri. |
| 104 | 35 | 65 | 62.6 | 32.5 | 4.9 | 1356 | 50 | β - C_3P , forst., peri. |
| | No. 72 | No. 24 | | | | | | |
| 151 | 90 | 10 | 73.5 | 16.9 | 9.6 | 1372 | 60 | peri., gl. |
| 152 | 80 | 20 | 72.0 | 19.5 | 8.5 | 1398 | 40 | Ng., peri., gl. |
| | | | | | | 1372 | 50 | Ng., peri., gl. |
| | | | | | | 1369 | 40 | β - C_3P , peri., forst., gl. |
| | | | | | | 1364 | 25 | no gl. |
| 153 | 70 | 30 | 70.5 | 22.0 | 7.5 | 1372 | 60 | peri., Ng., gl. |
| | | | | | | 1364 | 40 | peri., β - C_3P , forst., gl. |

temperature, were neglected in drawing this phase diagram at the step of this study. Detailed studies of the system containing nagelschmidtite will be made in future, because of the interesting scope of nagelschmidtite structure (super α -tricalcium phosphate by Nurse or its solid solution containing MgO and SiO_2).

System $3\text{CaO} \cdot \text{P}_2\text{O}_5$ -MgO-SiO₂.—The preliminary phase diagram, as shown in Fig. 6, was obtained from the results of previous paragraphs and the separate report of the system $3\text{CaO} \cdot \text{P}_2\text{O}_5$ -MgO-SiO₂-SiO₂¹⁾. The probable boundary curve between the phosphate and the forsterite fields in the $3\text{CaO} \cdot \text{P}_2\text{O}_5$ -2MgO-SiO₂-MgO-SiO₂ triangle was drawn from the results without a cooling experiment.

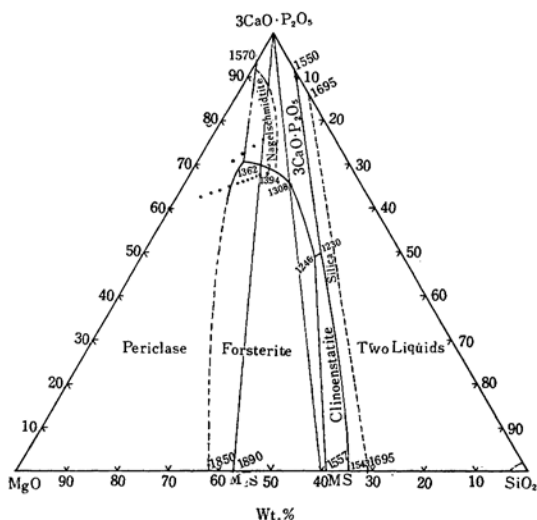


Fig. 6. The system $3\text{CaO} \cdot \text{P}_2\text{O}_5$ -MgO-SiO₂.

Summary

The phase relationship in the system $3\text{CaO} \cdot \text{P}_2\text{O}_5$ -MgO-SiO₂ was investigated by a cooling

experiment in the usual manner. The system $3\text{CaO} \cdot \text{P}_2\text{O}_5$ -MgO, a side of this triangle, is a simple binary one, its eutectic being located at the composition of 4~7% by weight of magnesia and $1570 \pm 20^\circ\text{C}$. The system $3\text{CaO} \cdot \text{P}_2\text{O}_5$ -2MgO-SiO₂ is not binary over the temperature of about 1350°C , below which it is considered as binary. Above 1350°C , nagelschmidtite is formed over the whole composition range and crystallized primarily from liquid at a high $3\text{CaO} \cdot \text{P}_2\text{O}_5$ portion. The minimum liquidus temperature is 1394°C , which is located at the composition of 70% C_3P and 30% M_2S . The system $3\text{CaO} \cdot \text{P}_2\text{O}_5$ -MgO-2MgO-SiO₂ was considered to be complex owing to the presence of nagelschmidtite, but the eutectic among their three components is located at the composition of 70% $3\text{CaO} \cdot \text{P}_2\text{O}_5$, 21% MgO, and 9% SiO₂, and $1362 \pm 5^\circ\text{C}$.

From the results described above and from those of the author's previous reports, the preliminary phase diagram of the system $3\text{CaO} \cdot \text{P}_2\text{O}_5$ -MgO-SiO₂ was constructed. In this diagram the problem that nagelschmidtite is formed in the composition range which is rich in $3\text{CaO} \cdot \text{P}_2\text{O}_5$ was not mentioned in detail. In addition to this, it was found by X-ray examination that β -tricalcium phosphate may be apparently stabilized at a higher temperature than β - α transition point, and has the appearance of dissolving magnesia component.

The author acknowledges the constant help and encouragement given by Professor R. Kiyoura, Tokyo Institute of Technology. This work was conducted as a part of a program to investigate the corrosion problems of various refractory bricks in the manufacture of fused phosphate fertilizer.

Research Laboratory of Engineering Materials
Tokyo Institute of Technology
Meguro-ku, Tokyo